

Al-Co-Fe (Aluminum-Cobalt-Iron)

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The phase relationships in this system were reviewed by [1988Ray], who presented two partial liquidus surfaces, an isothermal section at 800 °C for Co-rich alloys and isothermal sections at 640 and 600 °C for alloys near the Al corner. A full isothermal section at 650 °C was determined by [1999Koz] and updated by [2002Rag]. [2004Kam] determined the phase relationships in Co-rich alloys between 1200 and 900 °C in the region of magnetic and order-disorder transitions, briefly reviewed by [2005Rag]. [2008Rag] updated the results of [2004Koz] and [2006Koz], who used the liquid-quenching and composition-gradient methods to determine the metastable $B2/(B2 + A2)$ phase boundary at 650 °C in Co-rich alloys. Recently, [2010Gru] studied the phase equilibria of this ternary system at 1070 °C in the Al-rich region.

Binary Systems

In the Al-Co phase diagram [Massalski2], the intermediate phases in the Al-rich region are: Co_2Al_9 (monoclinic), $\text{Co}_4\text{Al}_{13}$, and Co_2Al_5 (hexagonal). Four crystalline forms occur around the composition $\text{Co}_4\text{Al}_{13}$: o- $\text{Co}_4\text{Al}_{13}$ (orthorhombic), m- $\text{Co}_4\text{Al}_{13}$ (monoclinic), Y- $\text{Co}_4\text{Al}_{13}$ (monoclinic), and Z- $\text{Co}_4\text{Al}_{13}$ (monoclinic) [2010Gru]. The structural characteristics of these forms were listed by [2010Gru]. In the Co-rich region, CoAl ($B2$, CsCl-type

cubic) occurs over a wide temperature and composition range (48-78 at.% Co). In the Fe-Al phase diagram [Massalski2], the solid solution based on the face-centered cubic (fcc) Fe is restricted by a γ loop. The solid solution based on the body-centered cubic (bcc) Fe exists in the disordered $A2$ and the ordered $B2$ and $D0_3$ forms. With increasing Al, four intermediate phases occur: ϵ ($D8_2$, Cu_5Zn_8 -type cubic), FeAl_2 (triclinic), Fe_2Al_5 (orthorhombic), and $\text{Fe}_4\text{Al}_{13}$ (monoclinic). In the Co-Fe system [Massalski2], a continuous fcc solid solution γ between fcc Fe and Co is stable over a wide temperature range. At 730 °C, the Fe-based bcc solid solution orders to a $B2$ structure via a second-order transition.

Ternary Isothermal Section

With starting metals of 99.999% Al, 99.99% Co, and 99.99% Fe, [2010Gru] levitation-induction melted about 20 Al-rich ternary alloys. The alloys were annealed at 1070 °C for 264 h. The phase equilibria were studied with x-ray powder diffraction and scanning electron microscopy. The phase compositions were measured by energy dispersive x-ray analysis. The isothermal section constructed by [2010Gru] at 1070 °C for Al-rich alloys is shown in Fig. 1. FeAl and CoAl form a continuous $B2$ solid solution. Also, the isostructural compounds $\text{Fe}_4\text{Al}_{13}$ and m- $\text{Co}_4\text{Al}_{13}$ form a continuous monoclinic solid solution. The other three modifications of $\text{Co}_4\text{Al}_{13}$ are also stable at this temperature [2010Gru]. Co_2Al_5 dissolves about 20 at.% Fe. [2010Gru] did a limited number of experiments at 800 °C and established the presence of Co_2Al_9 at this temperature. It dissolves up to ~5 at.% Fe and forms a tie-triangle with the Al-rich liquid and $(\text{Fe,Co})_4\text{Al}_{13}$.

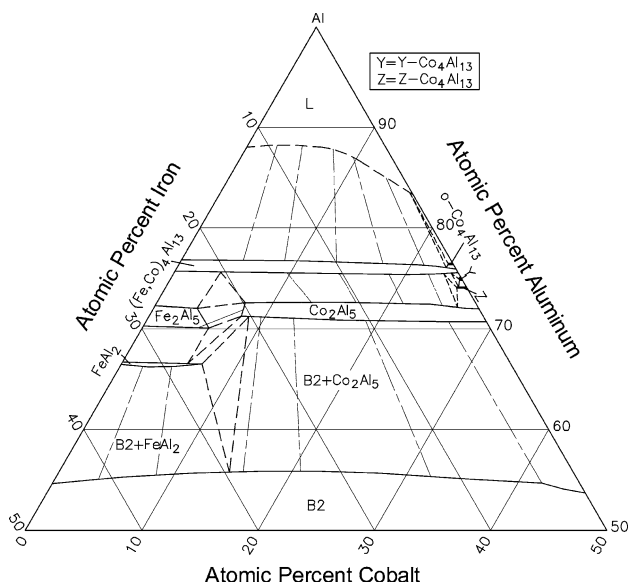


Fig. 1 Al-Co-Fe isothermal section at 1070 °C for Al-rich alloys [2010Gru]

References

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Section II: Phase Diagram Evaluations

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